

# Measurement of Helium Diffusion in Metals

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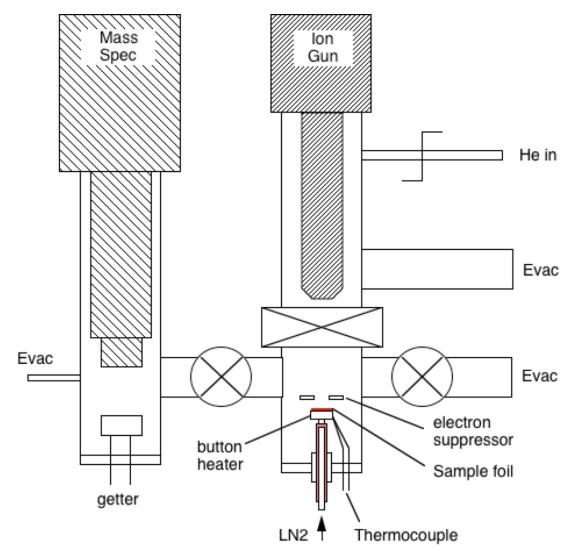


#### **Overview**

- Helium is generated in tritium exposed materials by beta decay.
- It is insoluble in metals and diffuses to the surface or collects in highpressure bubbles, eventually affecting the materials structural integrity.
- Modeling this process requires knowledge of the effective He diffusion and trapping parameters.
- Previous efforts with He implantation have focused on bubble growth and the onset of blistering.
- Here, a new, "gentle" implantation technique is described where He clustering is reduced. It involves
  - a low energy, short He implantation pulse at low temperature
  - followed by a rapid thermal desorption ramp.
- An analytic expression of desorption behavior yields both diffusion and trapping data.
  - Fickian diffusion from slab (short time limit)
  - SRIM implantation profile
- Results: first experimental data on He diffusion in Ni, Cu, and Pd.

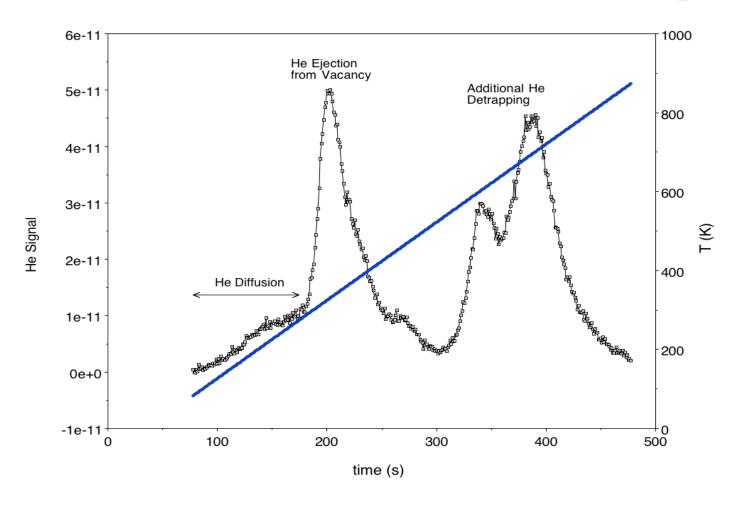
 $1809 \,\mathrm{dfc}$ 

## **Experimental System**



- Samples, 100 µm foils annealed: large grain
- He energy (0.5-5.0 keV)range = 4-20 nm
- Spot size, 4-6 mm
- Implant pulse (1-100 nC) .1-10 appm He
- Sample evacuation (20 s)
- Rapid He-TDS (1-2 K/s)

## The thermal desorption spectrum shows both diffusive behavior and He release from traps



## Analytic expression for diffusive desorption

• Diffusion from a plane sheet of half-thickness R (Crank, per Kass\*):

$$M_t = 2 (Dt/\pi R^2)^{1/2} \{ 1 + \sum \exp(-nR^2/Dt) \& \operatorname{erfc}(nR/\sqrt{Dt}) \}$$

For small times, the release from one side,  $M_t = (Dt/\pi R^2)^{1/2}$ 

 Crank also shows this solution for constant diffusivity D may be transformed for D(t) by replacing Dt by

$$\tau = \int_0^t D(t') dt'$$

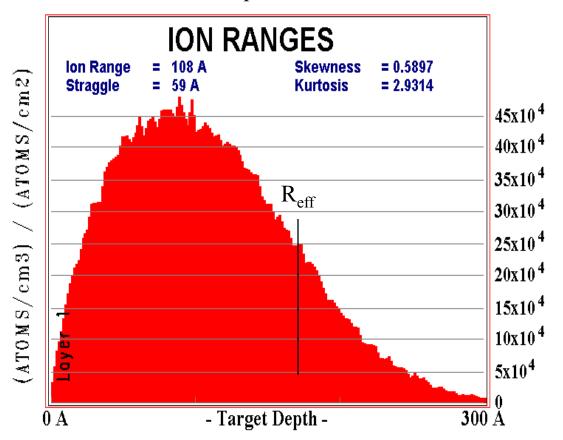
- With a linear ramp  $T = T_0 + \beta t$  and  $D = D_0 \exp(-E_D/kT)$ ,  $\tau = (D_0kT^2/\beta E_D) \exp(-E_D/kT)$
- Substituting, for large E<sub>D</sub>/kT

$$\begin{split} M_t &= (kT^2D_0/\pi\beta R^2E_D)^{1/2} \ exp \ (-E_D/2kT), \qquad \text{proportional to } 1/R \\ \text{or} \\ &\ln(M_t/T) = \frac{1}{2} \ln \left(kD_0/\pi\beta R^2E_D\right) + E_D/2kT, \quad \text{linear with } 1/T \end{split}$$

<sup>\*</sup>W.J. Kass, J. Vac. Sci. Technol. 14 (1977) 518.

### R is determined from Implantation Profile

SRIM: 2 keV He implantation into Pd



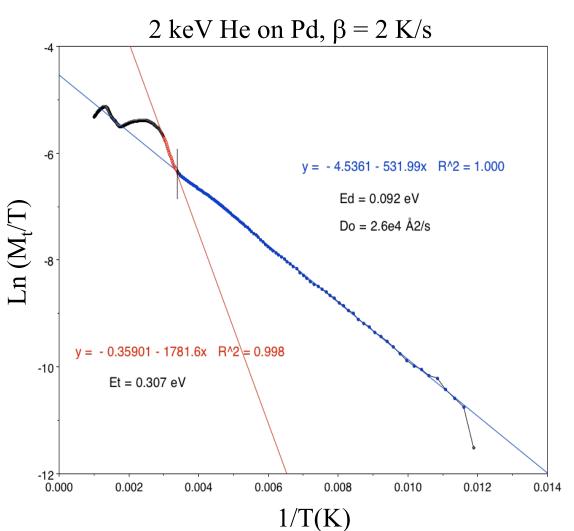
- For non-interacting He, the cumulative release is the sum of release from layers with R determined by *right* side of implant profile.
- Since  $M_t \propto 1/R$ , the effective thickness is

$$1/R_{eff} = (1/n) \sum_{n} (1/R_{n})$$

2 keV He on Pd:  $R_{eff} = 169 \text{ Å}$ 

• For a 500 s spectrum, errors from the absence of release from the *left* side of the profile (R<R<sub>eff</sub>/10) occur during the 20 s evacuation.

## Typical fit to plot of Ln $(M_t/T)$ vs 1/T



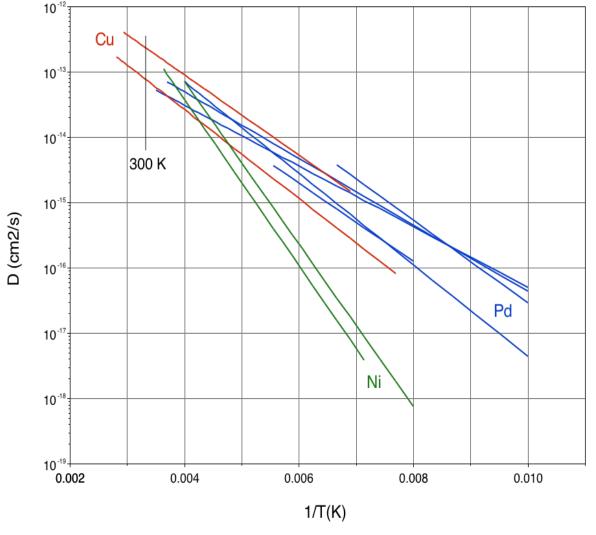
- Plot is linear over several orders.
  - From slope S,  $E_D = 2kS = 0.092 \text{ eV}$
  - From y-intercept I,  $D_0 = 2\pi\beta R^2 S \exp(2I)$   $= 2.6 \text{ e-}12 \text{ cm}^2/\text{s}$
- Rapid rise yields  $E_T = 0.32 \text{ eV}$

(Pd interstitial-vacancy recombination will eject He from vacancy)

## Table of results

Run	E(keV)	β(K/s)	E <sub>D</sub> (eV)	D <sub>0</sub> (cm <sup>2</sup> /s)	D <sub>300K</sub> (cm <sup>2</sup> /s)	E <sub>T</sub> (eV)
Pd4d	2	2	.101	5.4e-12	1.02e-13	.381
Pd4e	2	2	.092	2.2e-12	0.62e-13	.307
Pd4f	2	2	.139	4.5e-11	2.06e-13	.349
Pd4g	2	1	.125	5.9e-11	1.95e-13	.361
Pd4h	2	1	.119	7.9e-12	0.78e-13	.318
Cu1e	2	2	.135	1.4e-11	0.75e-13	.406
Cu1f	4	2	.122	2.6e-11	2.27e-13	.389
Ni1c	2	2	.247	6.9e-9	4.76e-13	1.06
Ni1d	4	2	.252	4.6e-9	2.65e-13	

## Arrhenius plot comparing He diffusivities



- Uncertainty: shift is due to normalization
- Release from traps obscures end of diffusive spectrum.
- Correct by stopping ramp before release from trap.

### **Summary**

An experimental technique is being developed that appears capable of measuring He diffusivities in metals.

- $10^{10}$  to  $10^{11}$  He atoms are implanted by a short, few keV He ion pulse at low temperature
- The sample chamber is quickly evacuated, then opened to a getterpumped gas analyzer, with some additional He pumping for fidelity of the He desorption behavior
- The sample temperature is rapidly ramped to 400 K, producing
  - (i) interstitial He diffusion from the sample and
  - (ii) escape of He from trapping sites at higher temperatures.
- The desorption spectrum is analyzed with a linear expression describing Fickian diffusion, under short time and  $E_{\rm D}/kT > 10$  approximations.
  - The analysis gives both  $D_0$  and  $E_D$ .
- He diffusion in Cu, Ni, and Pd is found to be around 10<sup>-13</sup> cm<sup>2</sup>/s at room temperature, but He trapping differs significantly.

### Next Steps

- Reduce uncertainties
  - Energy (background correction)
  - Normalization (stop ramp)
- Lower implant temperature
- Vary implant fluence (clustering, trapping)
- Examine diffusivity in other materials
  - NG, fusion, alloys