

# **AllnP-Based LEDs for Efficient Red and Amber Emission**

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## Background and Importance

3k<sub>B</sub>T

650

0.2

24

0.19 0.14

**Emission Wavelength (nm)** 

600

### Overview

Red and amber phosphide-based LEDs have efficiencies well below blue InGaN LEDs, in part due to fundamental limitations of the materials from which they are made. The efficiencies of these LEDs must be improved to enable next-generation color-mixed white LED architectures.

We are addressing the limitations of phosphide-based LEDs by modifying the materials that are used for the active and cladding layers. The objective of this project is to demonstrate the feasibility of fabricating red and amber LEDs from ordered/disordered Al, In, P heterostructures and to quantify the performance improvements this design can yield over incumbent phosphide LEDs.

Color./*	Wavelengtl	h Emitter Type	Material	Spectral Half Width	PCE (%) @ 25 ℃	PCE (%) @ 85 ℃	Hot/Cold Factor	
Blue	459 nm	Direct	Ga <sub>1-x</sub> In <sub>x</sub> N	20 nm	66	63	0.95	
Green	530 nm	Direct	Ga <sub>1-x</sub> In <sub>x</sub> N	30 nm	24	22	0.90	
Amber	595-605 nm	Phosphor-Converted	Ga <sub>1-x</sub> In <sub>x</sub> N/Phosphor	80 nm	23	18	0.80	Metho
Amber	585-595 nm	Direct	(Al <sub>x</sub> Ga <sub>1-x</sub> ) <sub>0.5</sub> In <sub>0.5</sub> P	20 nm	19	8	0.40	values
Dad	615 pm	Direct	$(A \mid C_{2}) \mid D$	20 nm	11	24		Dest-Kr

dology: Power Conversion Efficiency (PCE) at 25°C and hot/cold factors are reported for nown commercially-available devices in each



(Al<sub>2</sub>Ga<sub>12</sub>)<sub>0.5</sub>In<sub>0.5</sub>P LEDs suffer from internal intervalley transfer losses and external carrier leakage losses. These losses are fundamental to the

neu	0151111	Direct	$(AI_x Ga_{1-x})_{0.5} III_{0.5} F$	201111	44	24	0.55	color range; PCE values at 85°C are computed from
Deep Red	650-670 nm	Direct	(Al <sub>x</sub> Ga <sub>1-x</sub> ) <sub>0.5</sub> In <sub>0.5</sub> P	20 nm	49	39	0.80	the PCE @ 25 $^{\circ}$ and hot/cold factors.

(Al Ga<sub>1-x</sub>)<sub>0.5</sub>In<sub>0.5</sub>P material system and can only be overcome by using another material.

Direct bandgap Al, In, P has a higher direct-indirect bandgap energy crossover, which helps to reduce intervalley transfer. We combine Al, In, P with control of spontaneous atomic ordering to increase barriers to electron leakage to improve the efficiencies of red and amber LEDs.

## Al In<sub>1-x</sub>P Properties and Growth



causes the bandgap to decrease via a reduction in the conduction band edge energy.

Al<sub>1</sub>In<sub>1</sub>P emits at 590 nm a compositions ~ 36-43% Al, depending on the ordering. We demonstrated  $\Delta(\Gamma_{dis}-\Gamma_{ord})$  and  $\Delta(X-\Gamma)$  offsets > 140 meV and are moving the composition toward the targeted range.

Given the offsets demonstrated here and in the literature, Al<sub>v</sub>In<sub>1-v</sub>P devices, the barriers to electron loss at amber emission wavelengths are expected to be similar to those in  $(Al_Ga_{1-y})_{0.5}In_{0.5}P$  red LEDs.



Photoluminescence (PL) measurements as a function of temperature and excitation density provide information about electron loss pathways.

PL from the cladding layer is observed at low temperatures, giving an indication of the energy barrier offset.

Time-resolved PL shows two characteristics decay times associated with radiative and non-radiative recombination that can be used to extract information about the internal quantum efficiency.

We initially used the ratio of PL intensity at 295 K/10K to estimate the internal quantum efficiency of simple Al<sub>x</sub>In<sub>1-x</sub>P double heterostructures and compared to  $(Al_xGa_{1-x})_{0.5}In_{0.5}P$  samples with similar structures. We are continuing to improve our  $Al_{In_{1-v}}P$  material to increase luminescence.



#### **Metamorphic Growth**

#### **Growth Structure**



**Cross-Sectional Transmission Electron Microscopy (TEM) Image** 



 $Al_In_1 P$  (x < 0.43) has a larger lattice constant than GaAs.

To prevent threading dislocations, a metamorphic buffer layer is used to extend the lattice constant before  $Al_{x}In_{1-x}P$  growth.

This approach reduces the dislocation density to  $10^4 - 10^5$  cm<sup>-2</sup>.



## **Project Metrics and Milestones**

#### **Milestones and Progress Indicators**

- Maximize ordering in Al In  $_{1-\gamma}$  P ( $\lambda$  = 590 nm) to be  $\eta$  > 0.45.
- Demonstrate internal  $\Gamma$ -X offsets of 130 meV in ordered Al In<sub>1-Y</sub>P ( $\lambda$  = 590 nm).
- Demonstrate electron confinement potentials of 130 meV in ordered/disordered Al In<sub>1-x</sub>P heterostructures ( $\lambda$  =590 nm).

Demonstrate ordered Al<sub>x</sub>In<sub>1-x</sub>P ( $\lambda$  = 590 nm) with IQE = 75%.

Demonstrate  $(Al_xGa_{1-x})_{0.5}In_{0.5}P$  ( $\lambda = 590$  nm) with IQE = 65%.



#### **Electroluminescence**

Current-voltage measurements of double heterostructure devices indicate potential improvements in series resistance.



We are currently performing electroluminescence measurements as a function of temperature and drive current to determine how ordered/disordered Al In<sub>1</sub>, P LEDs differ from incumbent (Al<sub>x</sub>Ga<sub>1-x</sub>)<sub>0.5</sub>In<sub>0.5</sub>P devices in terms of emission efficiency.

#### **Current and Potential Limiting Factors**

10

1. Oxygen impurities

Deep traps associated with oxygen impurities increase non-radiative recombination. SIMS measurements indicate our devices have 10<sup>17</sup> cm<sup>-3</sup> oxygen impurities due to reactor-specific issues. While we cannot remedy the reactor design, we are looking into the use of higher V/III ratios to prevent oxygen inclusion.

#### 2. Dislocations

We have limited the threading dislocations in our devices to 10<sup>5</sup> cm<sup>-2</sup>. However, we find that the pattern of misfit dislocations within the buffer layer affects the luminescence. We are currently exploring whether this is due to strain, composition modulation or non-radiative recombination.

3. Ordered domains and composition modulation

Domain boundaries between ordered variants and composition modulation on a short scale are observed. We do not yet know the extent to which domain boundaries or composition modulation impact non-radiative recombination.

4. Multiple quantum wells

The devices we have tested to-date mainly consist of double heterostructures without multiple quantum wells. We will begin to add in optimized MQW structures to improve radiative recombination.

#### **Results Summary, Future Research and Considerations**

We have shown that we can achieve large bandgap energy shifts with ordering. Our experiments suggest that we can achieve up to 200 meV direct/indirect and active/clad conduction band offsets, producing higher barriers to electron loss than incumbent  $(Al_xGa_{1-x})_{0.5}P$  LEDs. We have also shown that we can metamorphically grow high quality Al<sub>1</sub>In<sub>1-v</sub>P heterostructures on GaAs substrates. Luminescence results indicate areas for improved performance by reducing non-radiative recombination sites and optimizing the device structure.

Going forward, we will continue to optimize the growth conditions to improve



#### Impact analysis.

Demonstrate Al In P heterostructure LEDs ( $\lambda = 590$  nm) with a MQW active layer that exhibit higher EQE than equivalent  $(Al_xGa_{1-x})_{0.5} 1ln_{0.49}$ P LEDs ( $\lambda = 590$  nm).

Characterize loss channels in Al In, P MQW structures. Evaluate the advantages and limitations of this approach.

#### the Al<sub>1</sub>In<sub>1-</sub>P material, incorporate multiple quantum well structures and analyze the potential of this approach to improve amber LED efficiency.



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